



On the photocatalytic properties of elongated TiO₂ nanoparticles for phenol degradation and Cr(VI) reduction

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ABSTRACT

Elongated TiO₂ nanoparticles with high aspect ratio have specific advantages in separation processes. In this study, TiO₂ nanofiber was prepared via a hydrothermal reaction, and TiO₂ nanoparticles with varied structural properties were obtained using hydrothermal and calcination post-treatments. Photocatalytic phenol degradation and Cr(VI) reduction over these catalysts was investigated. Results showed that hydrothermally prepared TiO₂ nanofiber consisted of titanate with high aspect ratio. Calcining the as-prepared TiO₂ nanofiber at 400 and 600 °C led to the crystalline phase transformation from titanate to TiO₂-B and to the presence of mixed crystalline phases composed of TiO₂-B and anatase. In contrast, hydrothermal post-treatment results in the generation of pure anatase TiO₂ nanoparticles. For photocatalytic phenol degradation, calcining the as-prepared TiO₂ nanofiber resulted in an enhanced catalytic activity, whereas TiO₂ nanoparticles obtained by calcination post-treatment showed lower catalytic activities for photocatalytic Cr(VI) reduction compared to as-prepared TiO₂ nanofiber. TiO₂ nanofiber modified using the hydrothermal post-treatment, however, exhibited the highest catalytic activity among TiO₂ nanoparticles examined for both photocatalytic phenol degradation and Cr(VI) reduction. Furthermore, TiO₂ nanoparticles obtained by hydrothermal post-treatment showed the best sedimentation efficiency, highlighting its prominent potential as a readily separable and recoverable photocatalyst.

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1. Introduction

Since the discovery of photocatalytic water splitting and degradation of organic contaminants [1,2], photocatalytic abatement of environmental pollutants has attracted considerable interest from both academic and industrial societies. In principle, the photocatalytic reaction is implemented by the excitation of electrons from the valence band to the conduction band of the semiconductor upon light irradiation, by which the excited electrons and holes can be used in reduction and oxidation reactions, respectively. Therefore, photocatalysis process is believed to be capable of eliminating organic or inorganic pollutants in water via an oxidation or reduction mechanism [3–5].

TiO₂ has been considered as one of the most practical candidates due to its low toxicity, high stability and prominent catalytic efficiency [5]. From the viewpoint of field applications, however, the large band gap of TiO₂ and the difficulty in catalyst recovery are recognized to be the major disadvantages of powdered TiO₂ [3,6]. Preparation of supported photocatalyst, film catalyst or magnetically separable photocatalyst has been attempted to circumvent the difficulty in catalyst recovery associated with powdered catalyst

[7–10], whereas the advantage in the separation of the specific photocatalysts is generally achieved at the expense of photocatalytic activity [6,11].

Elongated TiO₂ nanoparticles synthesized via an alkaline hydrothermal route have attracted great attention due to its high specific surface area and large aspect ratio [12–14]. In addition, TiO₂ nanoparticles with high aspect ratio exhibit prominent advantages in the sedimentation process over powdered catalysts, effectively facilitating solid–liquid separation and catalyst recovery [15–18]. It is noteworthy that the structures of the elongated TiO₂ nanoparticles vary with the synthesis methods and post-treatments, which may substantially impact their photocatalytic activities. In general, as-synthesized TiO₂ nanoparticles in titanate form show relatively low photocatalytic activities for pollutant degradation [18–20]. Modification of as-synthesized TiO₂ nanoparticles by calcination may enhance their photocatalytic activity. For example, Yu et al. [21] modified TiO₂ nanotube using calcination and found that the photocatalytic activity of the modified TiO₂ nanotube for acetone oxidation in air was closely linked to the calcination temperature. Alternatively, the structures of TiO₂ nanoparticles as well as their photocatalytic behaviors can be tuned via the hydrothermal post-treatment. Zhu et al. [15] converted titanate nanofiber into anatase nanofiber using a hydrothermal reaction and observed the substantially enhanced photocatalytic activity of anatase nanofiber for the degradation of sulforhodamine dye as compared to titanate

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nanofiber. In parallel, Yu and Xu [17] synthesized anatase TiO₂ nanorods by the hydrothermal conversion of TiO₂ nanofibers in acid solution and observed a higher photocatalytic activity for aqueous Rhodamine B dye as compared to P25.

It should be pointed out that the previous studies on the catalytic behaviors of elongated TiO₂ nanoparticles mainly focused on the photocatalytic degradation of organic pollutants via an oxidation mechanism. To our best knowledge, however, the properties of TiO₂ nanoparticles for photocatalytic reduction have been seldom addressed. In this study, TiO₂ nanofiber was prepared using the alkaline hydrothermal method and the structures of TiO₂ nanoparticles were tuned using either thermal or hydrothermal post-treatments. Photocatalytic phenol degradation and Cr(VI) reduction over the catalysts were investigated to systematically evaluate the impacts of their structural properties on the catalytic behaviors for photocatalytic oxidation and reduction. For comparison purpose, Degussa P25, a commercially available TiO₂ consisting of 80% anatase and 20% rutile, was also tested as a reference catalyst for photocatalytic phenol degradation and Cr(VI) reduction.

2. Experimental

2.1. Catalyst preparation

All chemicals were of analytical grade and were used without further purification. TiO₂ nanofiber was synthesized using the alkaline hydrothermal method and other TiO₂ nanoparticles with varied structural properties were prepared using different post-treatments with the as-prepared TiO₂ nanofiber as the precursor. For the preparation of TiO₂ nanofiber, 4 g TiO₂ powder (P25, Degussa) was added into 80 ml of 15 M NaOH solution and the suspension system was further stirred at room temperature for 2 h. The mixture was then transferred to a Teflon-lined stainless autoclave which was heated at 150 °C for 48 h. The precipitate was recovered by filtration, followed by repeated washing with distilled water. The solid particles were again suspended in 200 ml of distilled water with solution pH adjusted to 4.0 using 0.1 M HNO₃ solution, and the mixture was further stirred for 8 h. The resulting TiO₂ nanofiber, denoted as TiO₂-NF, was dried at 80 °C for 10 h. Different TiO₂ nanoparticles were prepared by treating the as-synthesized TiO₂-NF using either calcination or hydrothermal reaction. For thermal post-treatment, as-synthesized TiO₂-NF was calcined at 400 or 600 °C for 4 h. The resultant TiO₂ nanoparticles are referred to as TiO₂-NF-X, where X denotes the calcination temperature (°C). For hydrothermal treatment, 2.0 g of as-synthesized TiO₂-NF was added into 80 ml of distilled water and the solution pH was adjusted to 2.0 using 0.1 M HNO₃ solution. The mixture was transferred into a Teflon-lined stainless autoclave, which was subjected to hydrothermal treatment at 175 °C for 48 h. The solid particles were recovered by filtration, followed by repeated washing by distilled water and drying at 80 °C for 10 h. The resultant TiO₂ nanoparticle is designated to TiO₂-NF-H.

2.2. Material characterization

X-ray diffraction (XRD) patterns were collected in a Rigaku D/max-RA powder diffraction-meter using Cu K_α radiation. Transmission electron microscopy (TEM) images of the samples were collected on a Hitachi H-800 transmission electron microscope. The specific surface areas of the samples were determined by N₂ adsorption at -196 °C (77 K) on a Micrometrics ASAP 2020 apparatus. The Raman spectra of the samples were recorded on the JY HR-800 instrument. The UV-vis spectra of the samples were obtained in SHIMADZU UV-2401PC UV/vis spectrometer using BaSO₄ as a reference. The surface zeta potentials (ζ) of the samples were

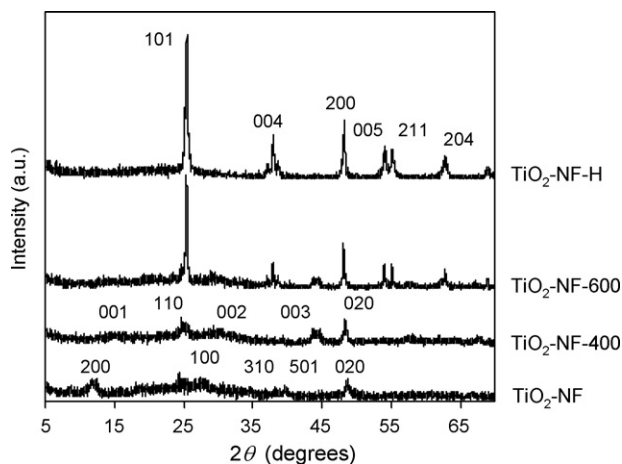


Fig. 1. XRD patterns of TiO₂ nanoparticles.

measured using a Zeta Potential Analyzer (Zeta PALS, Brookhaven Instruments Co.). Typically, 10 mg of the samples were dispersed in 100 ml of distilled water with varied pH and were equilibrated for 24 h prior to zeta potential measurements.

2.3. Photocatalytic reactions

Photocatalytic reactions were conducted in a commercial NDC photo-reactor equipped with a 500 ml cylindrical Pyrex vessel irradiated directly by a high pressure mercury lamp (125 W), which was described in Figure S1 (see Supporting information). For photocatalytic phenol degradation, 0.2 g of the catalyst was mixed with 500 ml of 50 mg l⁻¹ phenol solution which was further stirred in the dark for 60 min to reach the adsorption equilibrium prior to the photocatalytic test. During the photocatalytic reaction, samples were collected at selected time intervals and the catalyst particles were removed by centrifugation. The residual phenol concentration was determined using HPLC (Agilent 1200, USA), equipped with an ultraviolet (UV) detector and a C18 reversed phase column (250 mm × 4.5 mm, 5 μm, Agilent, USA) at 30 °C. The mobile phase consists of water and acetonitrile (40/60, v/v) with a flow rate of 1.0 ml min⁻¹.

As for photocatalytic Cr(VI) reduction, 0.5 g of the catalyst was suspended in 500 ml of 40 mg l⁻¹ Cr(VI) solution with solution pH 2.5 adjusted using 1.0 M H₂SO₄. The suspension was further stirred in the dark for 60 min. During the photo-reaction, samples were collected at preset time intervals. The catalyst powders were removed by filtration and the residual Cr(VI) concentration was determined spectrophotometrically using diphenylcarbazide as the color agent.

3. Results and discussion

3.1. Material characterization

The XRD patterns of the catalysts are compared in Fig. 1. For TiO₂ nanofiber, weak diffraction peaks were observed with 2θ at 12.4°, 28.4°, 40.1° and 48.7°, attributed to monoclinic titanate H₂Ti₃O₇·xH₂O [22,23]. The generation of titanate under alkaline hydrothermal conditions is based on a dissolution–recrystallization mechanism, by which TiO₂ precursor is irreversibly converted into soluble titanate species and these species are further thermodynamically deposited and crystallized into lamellar titanate [24,25]. Upon calcination of as-synthesized TiO₂ nanofiber at 400 °C for 4 h, new diffraction peaks were observed at 25.4°, 30.3°, 44.3° and 48.4°, assigned to metastable polymorph TiO₂-B [26,27]. This conversion could be ascribed to the dehydration and recrystallization

of layered titanate during the calcination process [12,28]. Further increasing the calcination temperature from 400 to 600 °C led to the emergence of new diffraction peaks, attributed to anatase TiO₂. It is noteworthy that the diffraction peaks characteristic of TiO₂-B still visible in the XRD pattern of TiO₂-NF-600, suggesting that TiO₂-NF-600 virtually consists of both TiO₂-B and anatase phases. In contrast, much different XRD pattern was observed upon hydrothermal post-treatment of as-synthesized TiO₂ nanofiber. As shown in Fig. 1, the diffraction peaks characteristic of typical anatase TiO₂ were identified. Moreover, other diffraction peaks characteristic of titanate, TiO₂-B and rutile were not observed, indicative of the presence of pure anatase TiO₂ crystalline phase. The crystalline conversion of titanate into anatase is generally ascribed to a topochemical reaction process, by which titanate is dehydrated under acidic conditions and in-situ converted into anatase with retention of its pristine morphology [16].

The crystalline phase transformation resulting from different post-treatments was also confirmed by Raman analysis. The Raman spectra of TiO₂ nanoparticles are presented in Fig. 2. For TiO₂-NF, the bands characteristic of titanium and oxygen stretching vibration in the TiO₆ octahedra were observed at 157.3, 190.8, 264.3, 361.4, 421.6, 457.5, 592.9 and 673.3 cm⁻¹, which are in good agreement with the Raman spectrum of H₂Ti₃O₇ obtained by Kolen'ko et al. [29]. Calcination of TiO₂-NF led to the disappearance of Raman peaks characteristic of titanate in concomitance of the presence of new Raman peaks at 160.8, 187.3, 413.4, 464.4 and 646.6 cm⁻¹, assigned to metastable TiO₂-B [27]. Further increasing the calcination temperature to 600 °C resulted in new Raman peaks at 144.9,

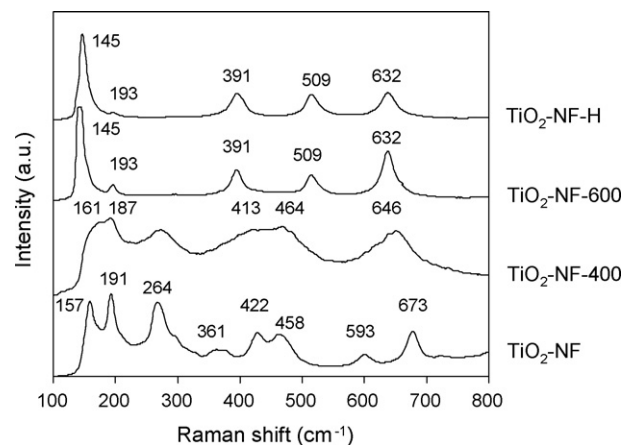


Fig. 2. Raman spectra of TiO₂ nanoparticles.

192.8, 390.7, 508.6 and 631.6 cm⁻¹, assigned to the active modes of typical anatase TiO₂ [30,31], suggesting the presence of anatase phase. For TiO₂-NF-H, similar Raman peaks characteristic of anatase TiO₂ were observed to that of TiO₂-NF-600, further indicating that TiO₂-NF-H consists of pure anatase phase.

The morphologies of TiO₂ nanoparticles can be clearly visualized using TEM and their TEM images are shown in Fig. 3. For as-synthesized TiO₂-NF, flexible TiO₂ nanofibers with diameters ranging from about 20–50 nm and length of several microm-

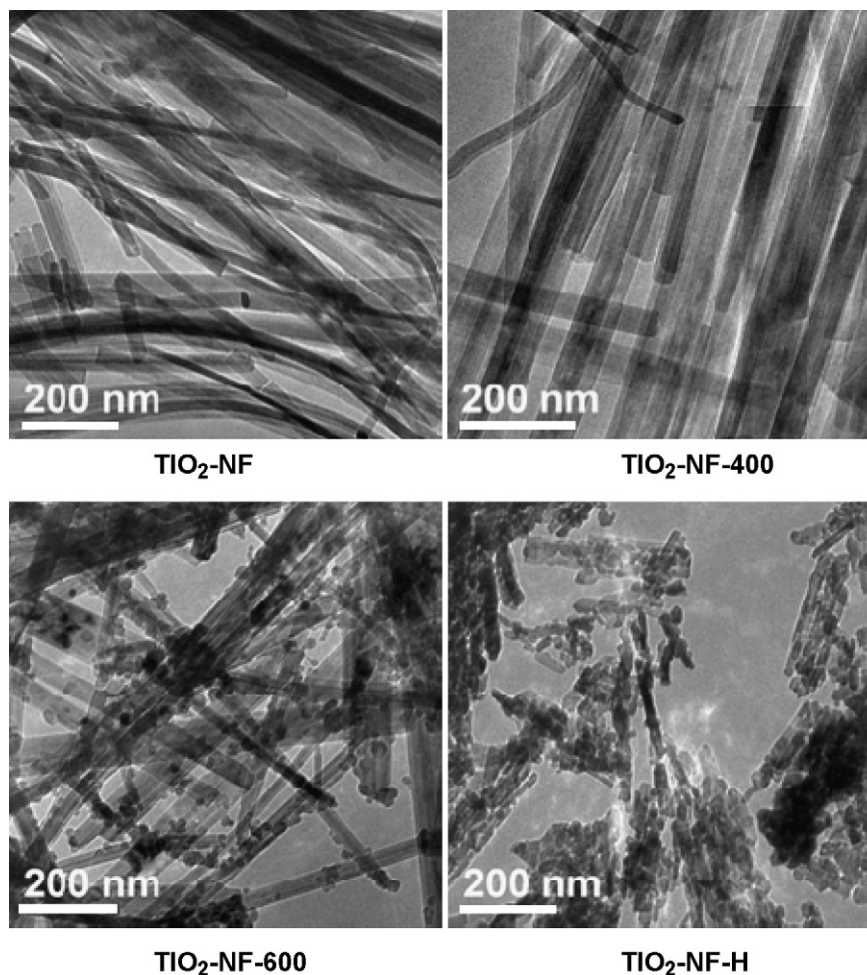


Fig. 3. TEM images of TiO₂ nanoparticles.

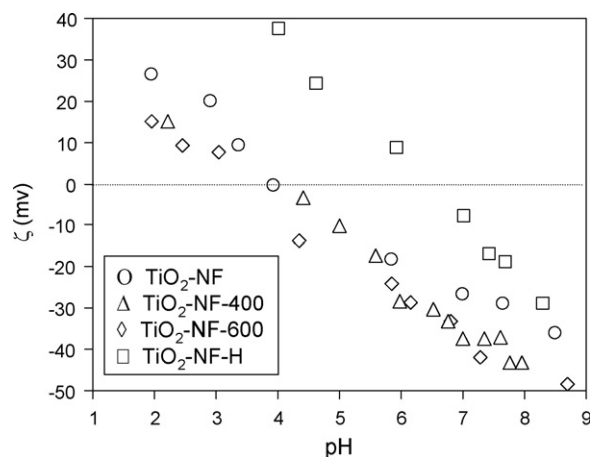


Fig. 4. Surface charge density (ζ) versus solution pH for different TiO₂ nanoparticles.

eters can be clearly observed, resulting from the fast growth of nanosheets under alkaline hydrothermal conditions [32]. For TiO₂-NF-400, minor changes were observed in comparison with as-synthesized TiO₂-NF. At calcination temperature of 600 °C, the TEM image of TiO₂-NF-600 displays rod-like morphology with markedly decreased diameters and the nanorods were decorated with a large amount of nanoparticles with diameters of about 10–20 nm, due to dehydrolyzation and crystallization of a fraction of titanate on the nanofiber surface. XRD results show that TiO₂-NF-600 consists of both TiO₂-B and anatase phases and TiO₂-NF-400 is composed of pure TiO₂-B. Therefore, it is speculated that the particles and nanorods consist of anatase and TiO₂-B, respectively. The TEM image of TiO₂-NF-H shows that the morphology of TiO₂-NF-H is much different from those obtained from thermal post-treatment. TiO₂-NF-H are intrinsically composed of nanoparticles with diameters around 10–20 nm which further aggregate into nanofibers.

Zeta potentials (ζ) of the nanoparticles as a function of solution pH is compiled in Fig. 4 and the obtained values of isoelectric points (IEPs) of the nanoparticles are listed in Table 1. For all TiO₂ nanoparticles, the surface charge densities monotonically decreased with the increase of solution pH. The IEPs were estimated to be 4.0, 4.0 and 3.9 for TiO₂-NF, TiO₂-NF-400 and TiO₂-NF-600, respectively. The identical IEP of TiO₂-NF-400 to that of TiO₂-NF reflects the similar nature of surface hydroxyl group of TiO₂-B to that of titanate. In addition, the similar IEP of TiO₂-NF-600 to TiO₂-NF-400 suggests that the surface of TiO₂-NF-600 is dominated by surface hydroxyl groups from TiO₂-B instead of anatase TiO₂ in TiO₂-NF-600 due to the substantially high IEP of anatase TiO₂ [33]. The IEP of TiO₂-NF-H was found to be about 6.5, identical to that of anatase [33].

The results of N₂ adsorption analysis are listed in Table 1. BET surface area of TiO₂-NF was found to be 55.6 m² g⁻¹. Calcination at 400 °C led to a slight decrease in BET surface area of TiO₂-NF-400, whereas further increasing the calcination temperature to 600 °C resulted in a sharp decrease of BET surface area to 24.6 m² g⁻¹, probably due to the transformation of a fraction of TiO₂ nanofiber into TiO₂ nanoparticles. For TiO₂-NF-H, BET surface area is 44.9 m² g⁻¹, slightly lower than those of TiO₂-NF and TiO₂-NF-400.

Table 1
Structural properties of TiO₂ nanoparticles.

Sample	Crystalline phase composition	BET (m ² g ⁻¹)	IEP	Band gap (eV)
TiO ₂ -NF	Titanate	55.6	4.0	3.60
TiO ₂ -NF-400	TiO ₂ -B	52.0	4.0	3.37
TiO ₂ -NF-600	TiO ₂ -B + anatase	24.6	3.9	3.32
TiO ₂ -NF-H	Anatase	44.9	6.5	3.24

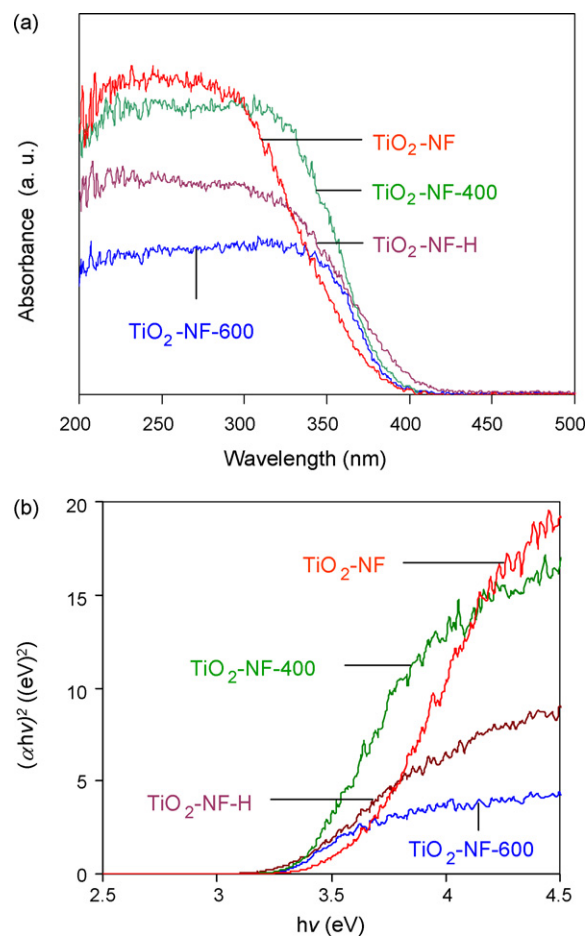


Fig. 5. (a) UV-vis spectra and (b) plots of $(\alpha hv)^2$ versus hv of the catalysts.

The UV-vis absorption spectra of the catalysts are shown in Fig. 5a. The absorption threshold of TiO₂-NF was slightly blue shifted, indicative of a larger band gap of TiO₂-NF as compared to other catalysts. For a direct semiconductor, the optical absorbance can be described using equation 1 [34]:

$$\alpha hv = B(hv - E_g)^{1/2} \quad (1)$$

where α is the absorbance intensity at light frequency of ν , B the absorbance constant and E_g the band gap energy, respectively.

The plots of $(\alpha hv)^2$ versus hv are compared in Fig. 5b. E_g is the intercept of linear extrapolation to hv axis and the obtained band gap energies for TiO₂-NF, TiO₂-NF-400, TiO₂-NF-600 and TiO₂-NF-H are found to be 3.60, 3.37, 3.32 and 3.24 eV, respectively, suggesting a considerably larger band gap of TiO₂-NF as compared to other TiO₂ nanoparticles tested in this study.

3.2. Photocatalytic phenol degradation

The photocatalytic degradation of aqueous phenol over the catalysts is compiled in Fig. 6. For phenol degradation over all photocatalysts, aqueous phenol concentration continuously dimin-

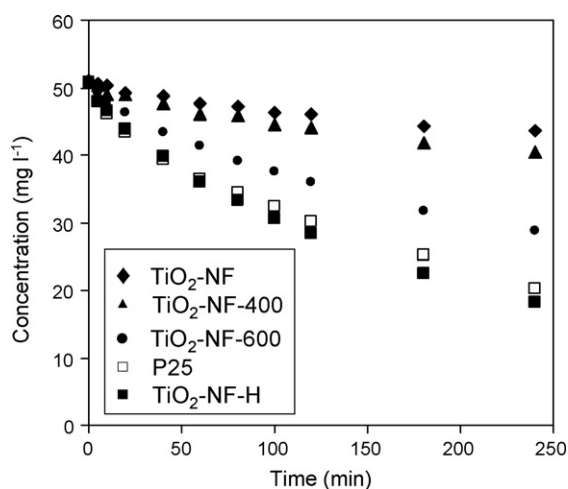


Fig. 6. Photocatalytic phenol degradation over TiO₂ nanoparticles.

ished with UV irradiation, indicating that the synthesized TiO₂ nanoparticles are effective photocatalysts. However, phenol degradation efficiency varied with the structural properties of the catalysts. After UV irradiation for 240 min, phenol removal were found to be 14.7%, 20.3%, 42.9%, 64.2% and 60.2% for TiO₂-NF, TiO₂-NF-400, TiO₂-NF-600, TiO₂-NF-H and P25, reflecting the highest photocatalytic activity of TiO₂-NF-H among the catalysts examined. As for degradation kinetics, the fitting results showed that phenol degradation over the photocatalysts could be well described using first-order kinetics:

$$r = -\frac{dC_t}{dt} = kC_t \quad (2)$$

$$\ln\left(\frac{C_0}{C_t}\right) = kt \quad (3)$$

where r is the reaction rate, C_t the concentration of aqueous phenol at reaction time t , C_0 the initial phenol concentration and k the reaction rate constant.

The fitting parameters of phenol degradation kinetics are listed in Table 2. The rate constants of photocatalytic phenol degradation were calculated to be 7.7×10^{-4} , 9.4×10^{-4} , 2.3×10^{-3} , 4.2×10^{-3} and $3.6 \times 10^{-3} \text{ mg l}^{-1} \text{ min}^{-1}$ for TiO₂-NF, TiO₂-NF-400, TiO₂-NF-600, TiO₂-NF-H and P25, respectively, indicative of an increasing catalytic activity order of TiO₂-NF < TiO₂-NF-400 < TiO₂-NF-600 < P25 < TiO₂-NF-H.

In principle, the photocatalytic activity is closely linked to the structural properties of the photocatalyst, such as crystallinity, crystalline phase composition and specific surface area. Because BET surface area of TiO₂-NF is approximately identical to that of TiO₂-NF-400, the higher phenol degradation rate of TiO₂-NF-400 compared to TiO₂-NF is attributed to the lower photocatalytic activity of its titanate structure [18–20]. Furthermore, the band gap of TiO₂-NF was found to be 3.60 eV, substantially larger than other catalysts. The larger band gap of TiO₂-NF leads to less effective absorption of excitation light, also giving rise to the low photocatalytic activity. Although the specific surface areas of TiO₂-NF-400

and TiO₂-NF are about two times as large as TiO₂-NF-600, substantially higher phenol degradation rate was observed for TiO₂-NF-600 than TiO₂-NF-400 and TiO₂-NF, suggesting that the higher phenol degradation efficiency of TiO₂-NF-600 results from the different crystalline phase composition of TiO₂-NF-600. In general, TiO₂ consisting of anatase exhibits higher photocatalytic activity as compared to TiO₂ with other crystalline phases [3–5]. Moreover, semiconductor containing mixed crystalline phases may show higher photocatalytic activity as a result of the enhanced separation and effective inhibition of recombination of excited electrons and holes [35,36]. Note that TiO₂-NF-600 consists of both TiO₂-B and anatase. In comparison with TiO₂-NF-400, therefore, the higher photocatalytic activity of TiO₂-NF-600 for phenol degradation can be attributed to either the presence of anatase or to the specific crystalline composition of mixed TiO₂-B and anatase.

As for TiO₂-NF-H, the phenol degradation was further enhanced in comparison with other catalysts tested. It should be pointed out that phenol degradation over TiO₂-NF-H is even faster than that over P25, a commercially available TiO₂ consisting of mixed anatase and rutile. Similarly, higher photocatalytic activities of TiO₂ nanoparticles prepared hydrothermally using TiO₂ nanotube as the precursor were also observed as compared to P25 [18–20]. Yu et al. [37] studied the photocatalytic acetone oxidation over TiO₂ nanofiber prepared from hydrothermal post-treatment of TiO₂ nanotube and attributed the enhanced activity to its smaller crystallite size, larger specific surface area, and higher pore volume. For TiO₂-NF-H, however, its BET surface area is approximately identical to those of P25, TiO₂-NF and TiO₂-NF-400, suggesting that other structural factors of TiO₂-NF-H account for the enhanced photocatalytic activity for phenol degradation. It is noteworthy that surface hydroxyl groups of TiO₂ are believed to be favorable for the generation of the highly active hydroxyl radical beneficial for oxidative degradation of organic compounds [38,39]. In contrast to other TiO₂ nanoparticles obtained using thermal post-treatment, TiO₂ nanoparticle prepared by the hydrothermal route is generally abundant in surface hydroxyl groups [20,40], which facilitates the generation of hydroxyl radical upon UV irradiation and eventually enhances phenol degradation.

For a heterogeneous photocatalytic system, in principle, the reaction presumably occurs on the catalyst surface. Therefore, reactant adsorption on catalyst surface is believed to be a prerequisite step and the photocatalytic reaction generally follows the Langmuir–Hinshelwood model. In order to verify the photocatalytic mechanism of phenol degradation over TiO₂-NF-H, the influence of initial phenol concentration on the initial degradation rate was conducted and the results are shown in Fig. 7. As shown in Fig. 7a, phenol degradation rate fast increased and then gently augmented with the increase of initial phenol concentration. The dependence of phenol degradation rate on the initial concentration was fitted according to the Langmuir–Hinshelwood model,

$$r_0 = -\frac{dC_0}{dt} = -k \frac{bC_0}{1 + bC_0} \quad (4)$$

$$\frac{1}{r_0} = \frac{1}{kb} \frac{1}{C_0} + \frac{1}{k} \quad (5)$$

Table 2
Rate constants of photocatalytic phenol degradation and Cr(VI) reduction.

Sample	Phenol degradation ($\text{mg l}^{-1} \text{ min}^{-1}$)	R^2	Cr(VI) reduction ($\text{mg l}^{-1} \text{ min}^{-1}$)	R^2
TiO ₂ -NF	7.7×10^{-4}	0.97	2.3×10^{-3}	0.98
TiO ₂ -NF-400	9.4×10^{-4}	0.97	1.5×10^{-3}	0.97
TiO ₂ -NF-600	2.3×10^{-3}	0.98	1.6×10^{-3}	0.98
TiO ₂ -NF-H	4.2×10^{-3}	0.99	7.3×10^{-3}	0.98
P25	3.6×10^{-3}	0.98	1.1×10^{-2}	0.97

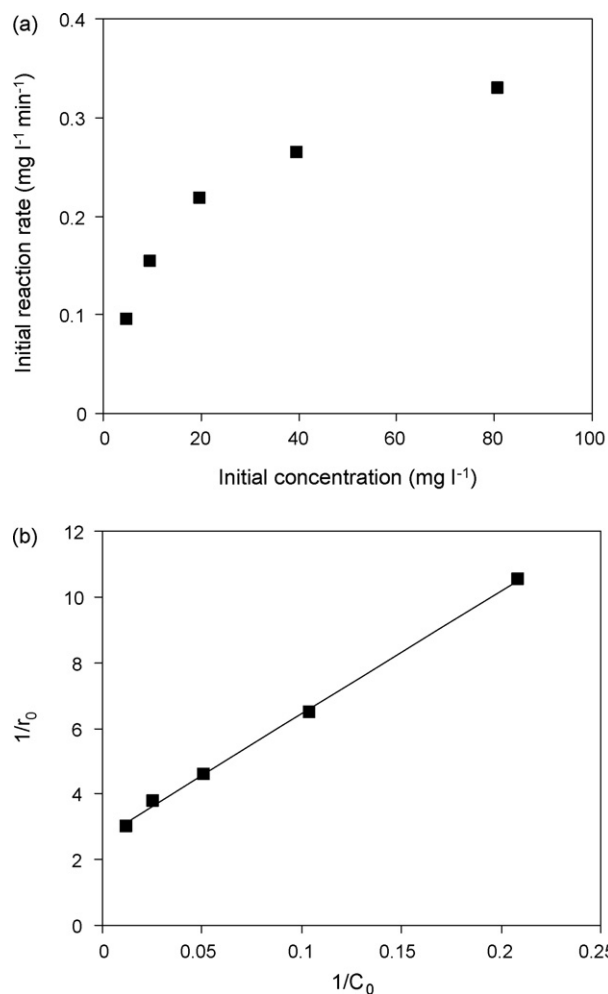


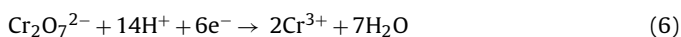
Fig. 7. (a) Dependence of initial phenol degradation rate on initial phenol concentration and (b) linear plot of $1/r_0$ versus $1/C_0$.

where r_0 is the initial phenol degradation rate at initial concentration C_0 , k the reaction rate constant and b the equilibrium constant for reactant adsorption on TiO_2 surface.

The plot of $1/r_0$ values versus $1/C_0$ is shown in Fig. 7b. The linear plot of $1/r_0$ versus $1/C_0$ with R^2 higher than 0.99 suggests that phenol degradation over TiO_2 -NF-H can be well described using Langmuir–Hinshelwood model, reflecting that phenol degradation over TiO_2 -NF-H is controlled by phenol pre-adsorption.

3.3. Photocatalytic Cr(VI) reduction

Besides photocatalytic phenol degradation, photocatalytic reduction of aqueous Cr(VI) was also preformed to evaluate the catalytic activities of TiO_2 nanoparticles. In the presence of photocatalyst, Cr(VI) can be reduced to Cr(III) by the excited electrons upon UV irradiation and the overall reaction of photocatalytic Cr(VI) reduction could be described as follows [41,42]:



Photocatalytic Cr(VI) reduction over the examined catalysts at pH 2.5 are compiled in Fig. 8. After reaction for 180 min, aqueous Cr(VI) was removed by about 32%, 23%, 26%, 75% and 88% for TiO_2 -NF, TiO_2 -NF-400, TiO_2 -NF-600, TiO_2 -NF-H and P25, respectively. Similarly, the photocatalytic Cr(VI) reduction could be well described using first-order kinetics and the fitting results are listed

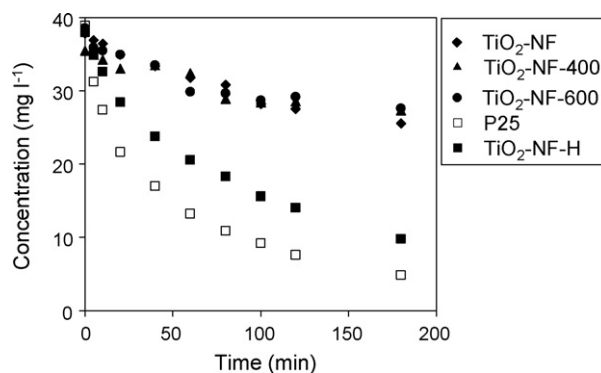


Fig. 8. Photocatalytic Cr(VI) reduction over TiO_2 nanoparticles.

in Table 2. The photocatalytic Cr(VI) reduction rate constants were calculated to be 2.3×10^{-3} , 1.5×10^{-3} , 1.6×10^{-3} , 7.3×10^{-3} and $1.1 \times 10^{-2} \text{ mg l}^{-1} \text{ min}^{-1}$ for TiO_2 -NF, TiO_2 -NF-400, TiO_2 -NF-600, TiO_2 -NF-H, reflecting an increasing catalytic activity order of TiO_2 -NF-400 \sim TiO_2 -NF-600 $<$ TiO_2 -NF $<$ TiO_2 -NF-H $<$ P25. Note that the photocatalytic activity of TiO_2 -NF is found to be higher than those of TiO_2 -NF-400 and TiO_2 -NF-600, and the Cr(VI) reduction rate constant of P25 is higher as compared to that of TiO_2 -NF-H, which is in contrast to the catalytic activity trend of photocatalytic phenol degradation.

For photocatalytic Cr(VI) reduction, Cr(VI) reduction efficiency is found to be pH dependent and a higher Cr(VI) removal can be

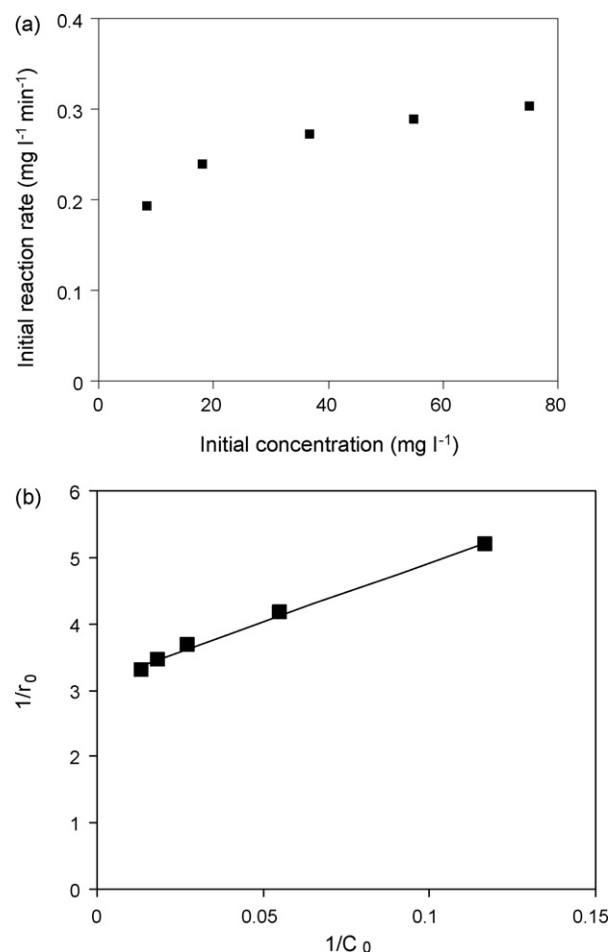


Fig. 9. (a) Dependence of initial Cr(VI) reduction rate on initial Cr(VI) concentration and (b) linear plot of $1/r_0$ versus $1/C_0$.

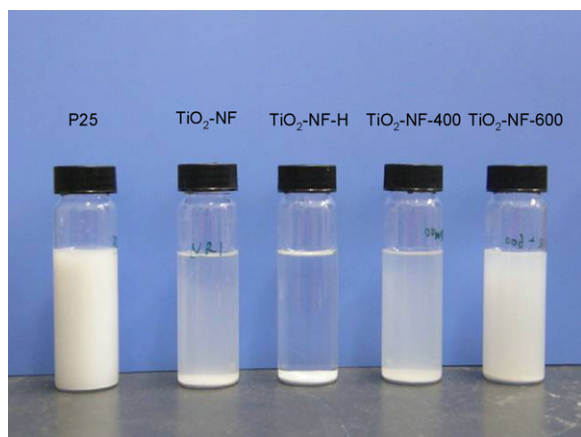


Fig. 10. Sedimentation properties of TiO₂ nanoparticles.

generally achieved at lower solution pH because the potential of CrO₄²⁻/Cr³⁺ pair decreases by 98 mV per pH unit with the increase of solution pH [43]. Therefore, the presence of acid sites on catalyst surface may favor Cr(VI) reduction. For example, Jiang et al. [44] compared the catalytic activities of alkaline treated sulfated TiO₂ and sulfated TiO₂ for photocatalytic Cr(VI) reduction and observed considerably higher photocatalytic activity of sulfated TiO₂ as compared to alkaline treated sulfated TiO₂ due to the decreased surface acidity of alkaline treated TiO₂. For titanate, Zheng et al. [45] studied the surface acidity of acid exchanged titanate using microcalorimetry with NH₃ as the probe molecule and detected a large amount of acid sites with varied acid strength. In parallel, TiO₂ nanotube was found to be active for the hydrolysis of 2-chlorethyl ethylsulfide due to the presence of surface acid sites [46]. Therefore, the considerably higher catalytic activity of TiO₂-NF than those of TiO₂-NF-400 and TiO₂-NF-600 can be attributed to the presence of surface acidity of TiO₂-NF although TiO₂-NF responds to relatively narrow irradiation light due to its large band gap and titanate is less active crystalline phase as indicated by photocatalytic phenol degradation. In comparison with TiO₂-NF-400, TiO₂-NF-600 shows a higher activity for phenol degradation, but comparable activity for Cr(VI) reduction, implying the more predominant impact of surface area than that of crystalline phase on photocatalytic Cr(VI) reduction. As for TiO₂-NF-H, it is understandable that similar to phenol degradation the highest catalytic activity for photocatalytic Cr(VI) reduction is achieved over TiO₂-NF-H among the examined nanofibers due to its anatase phase, comparable specific surface area with TiO₂-NF and TiO₂-NF-400, and considerable larger specific surface area than TiO₂-NF-600. It should be pointed out that in contrast to phenol degradation TiO₂-NF-H shows a lower catalytic activity for Cr(VI) reduction as compared to P25. For phenol degradation, the high surface hydroxyl group concentration accounts for the high catalytic activity of TiO₂-NF-H. Photocatalytic Cr(VI) reduction, however, is mediated by excited electrons instead of hydroxyl radicals. Therefore, the mixed anatase and rutile phase of P25 leads to its higher catalytic activity than TiO₂-NF-H for Cr(VI) reduction.

Similar to photocatalytic phenol degradation, photocatalytic Cr(VI) reduction also occurs on the surface of the photocatalyst, assumed to follow the Langmuir–Hinshelwood model. The influence of initial Cr(V) concentration on initial reaction rate was conducted and the results are shown in Fig. 9. The results were fitted according to equation 5. As shown in Fig. 9b, the linear plot of 1/r₀ versus 1/C₀ indicates that the photocatalytic Cr(VI) reduction can also be described using the Langmuir–Hinshelwood model, reflecting a surface adsorption controlled reaction mechanism.

The prominent advantages of TiO₂ nanoparticles with high aspect ratio are their excellent sedimentation properties. For the

tested TiO₂ nanoparticles, the sedimentation properties were preliminarily compared by suspending 25 mg of catalyst in 30 ml distilled water under stirring, followed by static sedimentation for 30 min and the results are shown in Fig. 10. Compared with P25, all synthesized catalysts were fast precipitated. Furthermore, TiO₂-NF-H exhibits the best sedimentation efficiency among the examined catalysts.

4. Conclusions

In this study, TiO₂ nanofiber was prepared via an alkaline hydrothermal reaction and TiO₂ nanoparticles with varied structural properties were obtained using thermal and hydrothermal post-treatments. Characterization results show that the crystalline phase of TiO₂-NF is titanate in nature. Thermal treatment of TiO₂-NF leads to the transformation of titanate to TiO₂-B at 400 °C and to mixed TiO₂-B and anatase at 600 °C, whereas pure anatase can be obtained using hydrothermal post-treatment. For photocatalytic phenol degradation, TiO₂-NF exhibits a low catalytic activity compared with the post-treated catalysts. In addition, the presence of mixed TiO₂-B and anatase in TiO₂-NF-600 considerably enhances phenol degradation. TiO₂-NF-H shows a higher catalytic activity than that of P25 due to its high concentration of surface hydroxyl groups. In contrast to photocatalytic phenol degradation, TiO₂-NF shows a higher catalytic activity as compared to TiO₂-NF-400 and TiO₂-NF-600 for photocatalytic Cr(VI) reduction, attributed to the presence of surface acidity in TiO₂-NF. The approximately identical catalytic activity of TiO₂-NF-400 to that of TiO₂-NF-600 suggests a more prominent impact of specific surface area on Cr(VI) reduction than that of crystalline phase composition. Because photocatalytic Cr(VI) reduction is mediated by excited electrons, TiO₂-NF-H has a relatively lower catalytic activity as compared to P25 for photocatalytic Cr(V) reduction.

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Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at doi:10.1016/j.jhazmat.2009.11.057.

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